

Metamagnetism of the manganese subsystem in RMn₂ intermetallic compounds

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The results of an investigation in the temperature range 5.5–300 K of the magnetic properties and crystal structure of RMn₂ intermetallic compounds having the C 15 cubic structure ($R = \text{Gd}, \text{Tb}$, or Dy) or the C 14 hexagonal structure ($R = \text{Ho}, \text{Er}$, or Tm) are presented. It is shown that whereas the transition to the magnetically ordered state is second order for $R = \text{Dy}, \text{Ho}, \text{Er}$, and Tm , for $R = \text{Gd}$ and Tb it is a first order transition with a discontinuous increase in the volume of the unit cell ($\Delta V/V \approx 15 \times 10^{-3}$ at 5.5 K). It is concluded that this behavior of GdMn_2 and TbMn_2 is related to the magnetic ordering of the manganese subsystem, which is of metamagnetic type. In the other RMn₂ compounds the exchange field acting on the manganese subsystem from the rare earth sublattice is weaker, so the manganese sublattice does not become ordered. The results are compared with the behavior of the 3d subsystem in similar RM₂ compounds with $M = \text{Fe}, \text{Co}$, and Ni .

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INTRODUCTION

The interest manifested in recent years in intermetallic compounds of the type RM₂, where R is a rare earth (RE) element and M is a 3d transition metal, is due to the fact that these compounds reveal a complex interaction of two magnetic subsystems—the RE and 3d-metal subsystems. Whereas the first subsystem is described by the localized-moment model, most investigators believe that the magnetism of the second subsystem is due to collectivization of the carriers.^{1,2} The essential feature is that, in a number of cases, the magnetization of the 3d subsystem exhibits metamagnetic behavior.^{3,4}

The ideas mentioned above were developed on the basis of studies of intermetallic compounds of RE elements with Fe, Co, or Ni. The RMn₂ compounds have been considerably less intensively studied. This is due mainly to technical difficulties that arise on account of the high vapor pressure of molten manganese. The comparatively few available studies^{5–8} of these compounds indicate that they are magnetically ordered at low temperatures. However, there is a wide spread between the values obtained in different studies for the magnetic ordering temperature, the saturation magnetic moments, etc., and the extent to which the ideas advanced concerning the nature of the magnetic ordering in RM₂ compounds with $M = \text{Ni}, \text{Co}$, and Fe can be extended to RMn₂ compounds has not been analyzed at all. To clarify this situation we have investigated the magnetic properties of the RMn₂ intermetallic compounds with $R = \text{Gd}, \text{Tb}, \text{Dy}$, and Ho , and have also investigated the crystal structure of these compounds and of ErMn_2 and TmMn_2 .

MEASUREMENT TECHNIQUE AND SPECIMENS

We measured the magnetizations and susceptibilities over the temperature range 4.2–350 K in the field of a superconducting solenoid at field strengths up to 60 kOe, using a vibration magnetometer. The measurement errors did not exceed 3%. The crystal structures were investigated over the

temperature range 5.5–300 K using a diffractometer with a low-temperature attachment. The measurements were made on polycrystalline specimens smelted in an electric arc or induction furnace in an argon atmosphere.⁹ The phase composition of the specimens was monitored by x-ray diffraction. Specimens intended for magnetic measurements were subjected, in addition, to a metallographic analysis. The impurity-phase concentration in the investigated specimens did not exceed 1%.

EXPERIMENTAL RESULTS

In accordance with the results of other studies,^{2,7} at room temperature the synthesized RMn₂ compounds with $R = \text{Gd}, \text{Tb}$, and Dy have a cubic crystal structure like that of the Laves phase of MgCu_2 (C 15), and those with $R = \text{Ho}, \text{Er}$, and Tm have a hexagonal structure like that of the Laves phase of MgZn_2 (C 14).

Figure 1 shows the temperature dependences of the magnetization and reciprocal susceptibility for several RMn₂ compounds. The susceptibility obeys the Curie-Weiss law at high temperatures. The paramagnetic Curie point Θ_p decreases on going from GdMn_2 to HoMn_2 (see Table I). It is also evident from this table that the experimental effective magnetic moments μ_{eff} of the investigated RMn₂ compounds are somewhat higher than those of the free R³⁺ ions; this may be attributed to the contribution of the manganese subsystem to the paramagnetic susceptibility.

Figure 2 shows the magnetization isotherms at 4.2 K of the investigated specimens. GdMn_2 , DyMn_2 , and HoMn_2 already reach saturation in 15–30-kOe fields, whereas TbMn_2 remains unsaturated even at 60 kOe.¹⁰ The fact that TbMn_2 remains unsaturated may be due to its high magnetic anisotropy; indeed, the fact that TbMn_2 , unlike the other investigated compounds, retains its maximum magnetization below T_c even in a 60-kOe field (Fig. 1) indicates that this may be the case.

TABLE I. Magnetic characteristics of RMn₂ inter-metallic compounds.

RMn ₂	$\mu_S(4.2^\circ\text{K})/\mu_B$	$\mu_{R\text{theor}}=gJ/\mu_B$	μ_{Mn}, μ_B	$\mu_{R\text{eff theor}}=\frac{g(J+1)H_z}{\mu}$	$\mu_{\text{eff theor}}/\mu_B$	$\Theta_p, ^\circ\text{K}$	$T_C, ^\circ\text{K}$	Order of the magnetic phase transition
GdMn ₂	2.7	7.0	2.15	7.94	8.4	100	110	I
TbMn ₂	~5.0 *	9.0	≤2.00	9.72	9.45	70	26	I
DyMn ₂	7.1	10.0	≤1.45	10.65	11.3	40	49	II
HoMn ₂	8.7	10.0	≤0.65	10.61	11.8	18	31	II

*In a 55 kOe field.

The saturation magnetic moments μ_s for the intermetallic compounds RMn₂ are given in Table I. The value given for TbMn₂ is only an estimate. It is evident from Table I that for all the investigated compounds μ_s is lower than the magnetic moment of the corresponding free RE ion. This difference is especially noticeable in the case of GdMn₂ and TbMn₂. Assuming that the nature of the magnetic interaction in RMn₂ compounds is the same as in other RM₂ compounds with heavy RE metals (a positive R-R exchange interaction leading to parallel orientation of the RE moments and a negative R-d interaction leading to antiparallel orientation of the RE and d moments¹), we find that the magnetic moment per manganese ion in RMn₂ compounds is

$$\mu_{\text{Mn}} = \frac{1}{2}(\mu_s - \mu_R). \quad (1)$$

For such calculations one must know the magnetic moment μ_R of the RE ion which, in the case of RE ions having a nonzero orbital angular momentum may be smaller than

$gJ\mu_B$ because of the effect of the crystal field. Only in the case of Gd, therefore, which is in an S state ($L = 0$) so that the effects of the crystal field can be neglected, is it possible to determine μ_{Mn} from the data obtained: $\mu_{\text{Mn}} = 2.15\mu_B$. For the other investigated intermetallic RMn₂ compounds we can only estimate μ_{Mn} from above (see Table I). Such an estimate is made more plausible by data of NMR studies,¹⁰ which yield the value $\mu_{\text{Mn}} = 0.4\mu_B$ for the cubic modification of HoMn₂, and by neutron diffraction data,¹¹ which yield $\mu_{\text{Mn}} \approx 0.8\mu_B$ in the cubic and hexagonal modifications of HoMn₂ (in the hexagonal modification, this is the average moment of manganese in inequivalent positions).

We used the method of thermodynamic coefficients¹² to determine the magnetic transition temperatures for the intermetallic compounds DyMn₂ and HoMn₂. The resulting data are given in Table I. It follows from our measurements, however, that the magnetization isotherms for GdMn₂ and TbMn₂ are not represented near the transition point by the formula $H/\mu = \alpha + \beta\mu^2$, which follows from the thermodynamic theory of second order magnetic phase transitions. For these compounds we used susceptibility measurements in weak fields (~ 30 Oe) to determine T_C . It is evident from Fig. 3 that the magnetic transition in GdMn₂ takes place at ~ 110 K and exhibits considerable hysteresis. This, as well as crystal structure data (see below), indicates that the magnetic transition in this compound is of first order. Contradictory data on the magnetic properties of GdMn₂ have been published in recent years; in particular, the Curie tempera-

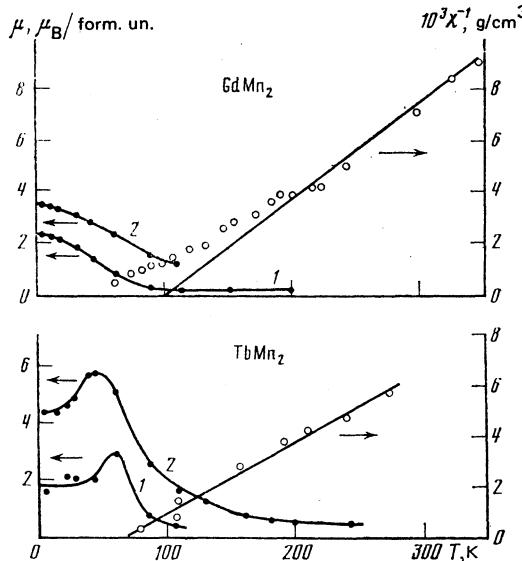


FIG. 1. Temperature dependences of the reciprocal susceptibility χ^{-1} and magnetization μ for GdMn₂ in 5 kOe (1) and 40 kOe (2) fields and for TbMn₂ in 10 kOe (1) and 40 kOe (2) fields.

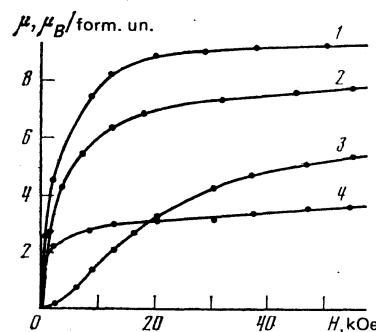


FIG. 2. Magnetization isotherms for RMn₂ compounds at 4.2 K: 1—HoMn₂, 2—DyMn₂, 3—TbMn₂, 4—GdMn₂.

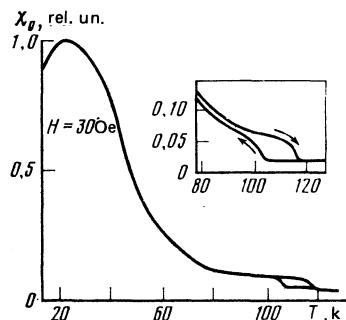


FIG. 3. Initial susceptibility χ_0 for GdMn_2 . The inset shows the transition region on a larger scale.

ture is given as 38, 86, 185, and 283 K in Refs. 7, 2, 13, and 8, respectively. A discontinuity in the initial susceptibility at 110 K was noted in Ref. 8, but this temperature was not interpreted as the transition point to the magnetically ordered state. The contradictory data might be due to the presence of small quantities of strongly magnetic impurities (for example, Gd and intermetallic compounds of it that differ stoichiometrically from GdMn_2). Some of our data also suggest this interpretation: we observed a small spontaneous magnetization even above $T_C = 110$ K in some specimens. Errors due to inaccuracies in determining T_C from data on temperature dependences of the magnetization in a field, as was done in Refs. 7 and 13, are also possible.

It follows from measurements of the temperature dependence of the susceptibility of TbMn_2 in weak fields that the magnetic transition in this compound is also a first order phase transition and takes place at 26 K.²⁾ This is also confirmed by crystal-structure data (see below). That the phase transition at T_C in TbMn_2 is of first order is also indicated by the unusual behavior of the magnetization above the transition point in the temperature range 26–50 K. Figure 4 shows typical $\mu(H)$ curves for this compound at various temperatures in the ordered region (below 25 K), the transition region (26–50 K), and the paramagnetic region (above 50 K). In weak fields up to 10 kOe the magnetization in the transition

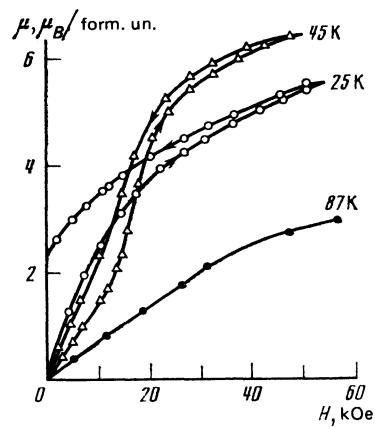


FIG. 4. Magnetization curves for TbMn_2 in the ordered (25 K), metamagnetic (45 K); and paramagnetic (87 K) regions.

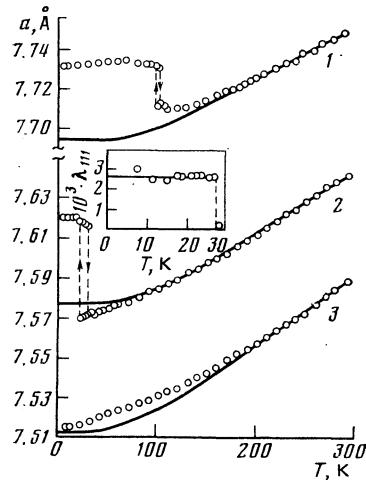


FIG. 5. Temperature dependence of the lattice constant a for cubic RMn_2 compounds: 1— GdMn_2 , 2— TbMn_2 , 3— DyMn_2 . The points represent experimental data and the curves show the phonon contribution. The inset shows the temperature dependence of the anisotropic magnetostriction constant λ_{111} for TbMn_2 .

region varies linearly with the field strength. When the field strength is further increased, the magnetization rises sharply. Hysteresis is also observed on the $\mu(H)$ curve. All this indicates that the magnetization of TbMn_2 is of metamagnetic type in the temperature region under discussion.

The data from the magnetic measurements do not enable us to draw final conclusions concerning the nature of the ordering of the manganese subsystem in the RMn_2 compounds. Crystal-structure studies of these compounds at various temperatures can yield useful information.

Figure 5 shows the temperature dependences of the lattice constants of the cubic RMn_2 compounds ($\text{R} = \text{Gd}, \text{Tb}$, and Dy). Here the full curves represent the theoretical dependences in the Debye theory approximation without allowance for the magnetic interactions (the phonon contribution to the thermal expansion). The curves were calculated for the Debye temperature $\Theta_D = 270$ K.⁶ In the high-temperature region the $a(T)$ curve has the normal trend, i.e., it falls monotonically with decreasing temperature. However, in GdMn_2 and TbMn_2 there is observed a discontin-

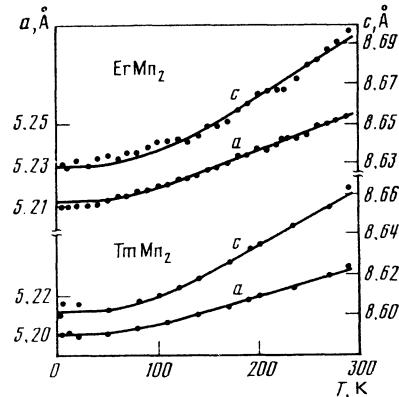


FIG. 6. Temperature dependence of the lattice constants a and c for hexagonal RMn_2 compounds. The points represent experimental data and the curves show the phonon contribution.

TABLE II. Structural characteristics of RMn₂ compounds.

RMn ₂	Structure	<i>a</i> , Å	<i>c</i> , Å	$\frac{\Delta V}{V}$	$\alpha_a \cdot 10^4$	$\alpha_c \cdot 10^4$
					at 290 K	
GdMn ₂	Cubic	7.748	—	$14.7 \cdot 10^{-3}$	34	—
TbMn ₂	»	7.648	—	$15.3 \cdot 10^{-3}$	40	—
DyMn ₂	»	7.590	—	$< 10^{-4}$	47	—
HoMn ₂	Hexagonal	5.330	8.600	$< 10^{-4}$	43	36
ErMn ₂	»	5.255	8.695	$< 10^{-4}$	37	36
TmMn ₂	»	5.238	8.660	$< 10^{-4}$	29	33

α_a and α_c are the thermal expansion coefficients along the *a* and *c* axes

uous decrease in the volume of the unit cell ($\Delta V/V \approx 15 \times 10^{-3}$) near the magnetic ordering temperature, which indicates that in these compounds the magnetic transition is first order.

In addition, the magnetically ordered phase of TbMn₂ exhibits a rhombohedral distortion of the crystal structure. The magnitude ε of this distortion (the deviation from 90° of the angle between edges of a cube) amounts to 2.8×10^{-3} rad at 8 K. As has been repeatedly shown,¹⁴⁻¹⁶ such distortions arise in RE laves phases as a result of strong anisotropic magnetoelastic interactions of single-ion nature and lead to deformation of the crystal lattice along the easy magnetization axis. In the case of TbMn₂, rhombohedral distortion (deformation along the ⟨111⟩ diagonal of the cube) indicates that the ⟨111⟩ direction is that of the easy axis. The inset in Fig. 5 shows the temperature dependence of the anisotropic magnetostriction constant λ_{111} for TbMn₂ in the magnetically ordered region. In accordance with the nature of the magnetic phase transition, it goes discontinuously to zero at T_c .

Unlike GdMn₂ and TbMn₂, DyMn₂ exhibited no appreciable bulk anomalies of the crystal structure on passing through the Curie point. It should also be noted that there are no distortions of the cubic structures of GdMn₂ and DyMn₂ in the magnetically ordered region; this indicates that the anisotropic magnetostriction of these compounds is comparatively weak ($< 10^{-4}$).

The lattice constants of hexagonal HoMn₂, ErMn₂, and TmMn₂ also exhibit no anomalies of any sort in the investigated temperature interval. Figure 6 shows the results for ErMn₂ and TmMn₂, whose Curie points, as determined from neutron diffraction measurements, are 25 and 12 K, respectively.¹⁷ The phonon contribution was calculated at $\Theta_D = 270$ K for the lattice constant *a* and at $\Theta_D = 350$ K for the lattice constant *c*. The lattice constants of HoMn₂ behave in a similar manner as the temperature is varied.

Some of the data from x-ray measurements of the RMn₂ compounds are summarized in Table II. The thermal expansion coefficients α for the RMn₂ compounds are three to four times larger than those for other RM₂ compounds (with M = Ni or Co)^{9,16}; this shows how unstable the crystal structure of the RMn₂ compounds is. The thermal expansion coefficient α reaches its largest value at the transition from the C 15 to the C 14 structure (between DyMn₂ and HoMn₂). This

leads to the instability of the crystal structure of HoMn₂, which exists in two modifications: hexagonal and cubic.² The deviation of the temperature dependence of the lattice constant of DyMn₂ from the Debye law, which can be interpreted as a temperature dependence of Θ_D , is also associated with the low stability of the crystal structure of that compound.

DISCUSSION

Thus, our magnetic and x-ray studies have shown that the RMn₂ compounds can be separated into two groups. In one of the groups (comprising GdMn₂ and TbMn₂) the transition to the magnetically ordered state is as a first order phase transition and is accompanied by gigantic magnetic bulk anomalies. In the other group (DyMn₂, HoMn₂, ErMn₂, and TmMn₂) the magnetic bulk anomalies are considerably smaller and the transition at the Curie point is first order. The observed features of the magnetic ordering of RMn₂ compounds have much in common with the properties of the RM₂ intermetallic compounds with M = Fe, Co, or Ni. Thus, in the RCo₂ series the magnetic transition is first order and is also accompanied by considerable ($\sim 5 \times 10^{-3}$) bulk anomalies.^{3,9,18} The $\mu(H)$ curves for these compounds are of metamagnetic type above the Curie point.¹⁹

As was already noted, the magnetic behavior of the RM₂ compounds can be well described under the assumption that two magnetic subsystems, one formed by the localized moments of the 4*f* electrons of the rare earths (localized RE magnetism) and the other consisting of a magnetic moment of band origin produced by the 3*d* electrons of the transition metal (*d* band magnetism), coexist within them.³ In Laves phases the magnetic moments of the rare earths are always ordered at low temperatures, whereas the magnetic moment of the *d* subsystem depends on structural features and on the population of the *d* band formed by the partial overlap of the 3*d* band of the transition metal, which is comparatively narrow (~ 4 eV) and has a high density of states, with the higher-energy 5*d* band of the RE metals, which is broader (~ 7 eV) and has a low density of states.⁴ Figure 7 shows a schematic density-of-states curve $N(\epsilon)$ for the *d* band of RM₂ compounds together with the position of the Fermi level for ferromagnetic, metamagnetic, and paramagnetic behavior of the *d* subsystem.

As is well known, according to the band model the Stoner criterion

$$IN(\epsilon_F) \geq 1 \quad (2)$$

(where I is the exchange interaction integral for the band d electrons and $N(\epsilon_F)$ is the density of states at the Fermi level) must be satisfied in order for ferromagnetism to arise. If $N(\epsilon_F)$ is large enough (position 1 in Fig. 7) the Stoner criterion will be satisfied as a result of the d - d exchange interaction, the $3d$ subsystem will be spontaneously magnetized, and the effective field H_{eff} from the RE subsystem will increase the magnetic moment only slightly. According to Ref. 4, such a situation obtains in the RFe_2 compounds (see curve 1 in Fig. 8).

In the RNi_2 compounds the d band is almost completely filled (position 2 in Fig. 7), the density of states at the Fermi level is low, and magnetic ordering of the d electrons does not take place in the field H_{eff} (curve 2 in Fig. 8).

The intermediate case, which is realized in the RCO_2 compounds,⁴ is the most interesting. In these compounds the Fermi level of the d electrons lies on the steeply falling part of the $N(\epsilon)$ curve (case 3 in Fig. 7), the density of states at the Fermi level is higher than in RNi_2 , but the Stoner criterion for d - d interaction is not satisfied ($IN(\epsilon_F) \approx 0.85$).⁴ The compounds YCo_2 and LuCo_2 , which contain nonmagnetic elements, are therefore exchange-enhanced Pauli paramagnets. In RCO_2 compounds containing magnetic rare earths an additional effective field H_{eff} due to the f - d exchange interaction acts on the cobalt subsystem. This reduces the energy of the subband spin-up (along the field) electrons and increases that of the spin-down (against the field) electrons (see the inset in Fig. 7). Then since the Fermi level lies on the falling part of the $N(\epsilon)$ curve ($N'(\epsilon_F) < 0$ and $N''(\epsilon_F) > 0$), the average density of states $N(\epsilon_F) = (N_+(\epsilon_F) + N_-(\epsilon_F))/2$ increases with increasing H_{eff} (see the inset in Fig. 7). There is a critical field strength H_{cr} above which the Stoner criterion is satisfied and the d subsystem undergoes a discontinuous transition to the magnetically ordered state, the magnetic moment of the d subsystem behaving in a metamagnetic manner as shown by curve 3 in Fig. 8.

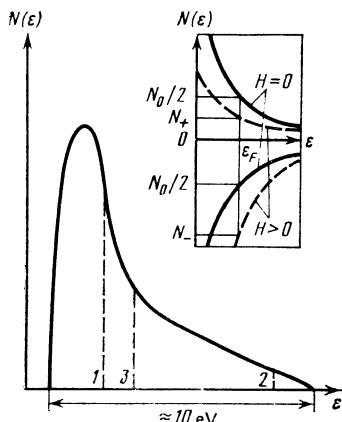


FIG. 7. Schematic density of states curve $N(\epsilon)$ for the d subsystem of RM_2 compounds, and the position of the Fermi level: 1— $M = \text{Fe}$, ferromagnetic state; 2— $M = \text{Ni}$, paramagnetic state; 3— $M = \text{Co}$, metamagnetic state. The inset shows the change in $N(\epsilon_F)$ when a field is applied.

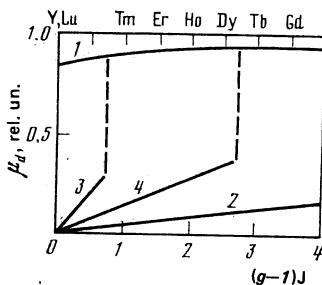


FIG. 8. Schematic depiction of the dependence on $(g - 1)J \propto H_{\text{eff}}$ of the magnetic moment μ_d of the $3d$ subsystem in RM_2 compounds: 1— $M = \text{Fe}$, 2— $M = \text{Ni}$, 3— $M = \text{Co}$, 4— $M = \text{Mn}$.

As a result of the metamagnetic character of the magnetization of the d subsystem, the transition to the magnetically ordered state at the Curie point in the RM_2 compounds becomes a first order phase transition. This situation is realized in the RCO_2 compounds with $R = \text{Dy}, \text{Ho}, \text{Er}$, and, apparently, Tm .³ We note that, as calculations show, a first order phase transition arises when the R- d exchange interaction is dominant and the R-R interaction is weak. In the opposite case the phase transition to the magnetically ordered state becomes second order. This may be one reason why the transitions to the magnetically ordered state in GdCo_2 and TbCo_2 are second order, for Gd and Tb have the strongest R-R exchange interactions of all the rare earths.

Under the assumption of the spin character of the f - d exchange, the effective field acting on the d subsystem is related to the g factor and principal quantum number J of the rare earth by the formula

$$H_{\text{eff}} \propto (g - 1)J \quad (3)$$

and increases on going from Tm to Gd . $H_{\text{eff}} > H_{\text{cr}}$ in the RCO_2 compounds—even in TmCo_2 —so the cobalt subsystem is magnetically ordered in all the RCO_2 compounds with heavy rare earths R (curve 3 in Fig. 8).

The d subsystem band metamagnetism model is also useful in discussing the magnetic properties of the RMn_2 compounds. In all cases magnetic ordering of the d subsystem in R- $3d$ intermetallides is accompanied by gigantic volume effects.^{9,18} Moreover, ordering of the R subsystem alone does not give rise to such anomalies.^{16,20} Among the RMn_2 compounds volume anomalies are observed only in those containing Tb or Gd, and one may conclude that the manganese subsystem is ordered only in TbMn_2 and GdMn_2 in which, according to (3), the effective field acting from the RE sublattice is the strongest. In the other RMn_2 compounds the effective field is weaker than H_{cr} and is incapable of ordering the manganese subsystem (curve 4 in Fig. 8).

Let us estimate H_{cr} for the RMn_2 compounds, assuming that the structure of the $3d$ band is the same in all these compounds. When $H < H_{\text{cr}}$ we have, according to the model under consideration,

$$\mu_{\text{Mn}} = \chi_{\text{Mn}} H_{\text{eff}}, \quad (4)$$

where χ_{Mn} is the paramagnetic susceptibility of YMn_2 and is equal to $1.28 \times 10^{-3} \text{ cm}^3/\text{mole}$ at 0 K.²¹ If we assume that $\mu_{\text{Mn}} = 0.6\mu_B$ in HoMn_2 (this is the average value from NMR

data and neutron diffraction studies and is in agreement with the results of magnetic measurements) we find that $H_{\text{eff}}^{\text{Ho}} = 4.2 \times 10^6$ Oe. Using this value, we can estimate H_{eff} for the other RMn_2 compounds from Eq. (3). It follows from our data (Fig. 8) that $H_{\text{eff}}^{\text{Dy}} < H_{\text{cr}} < H_{\text{eff}}^{\text{Tb}}$, whence $5.25 \times 10^6 < H_{\text{cr}} < 6.3 \times 10^6$ Oe; thus, H_{cr} is considerably larger for the RMn_2 compounds than for the RCO_2 compounds where⁴ $H_{\text{cr}} = 1 \times 10^6$ Oe.

Thus, the model of the band metamagnetism of the manganese subsystem enables us to describe many features of the magnetic properties of various RMn_2 compounds from a unified point of view and to compare them with the properties of other RM_2 compounds. That is the principal advantage of this approach.

It should be noted, however, that this model is still hypothetical and, although there are many indirect indications of its validity, it has not been confirmed by direct experiments. In addition, there are data that cannot be fitted into such a simple model. These include the information in one of the earlier studies²² on the noncolinear magnetic structure of TbMn_2 , the problem of the nature of the recently discovered²³ low-temperature structure transformation in YMn_2 , etc.

¹¹We do not give the hysteresis loops of the investigated specimens here, but only note that they depend on the magnetic history of the specimen. This is specially pronounced in TbMn_2 , as was noted earlier in Ref. 5. In Fig. 2 we give the magnetization curve of TbMn_2 for the first application of the field.

²²As our studies showed, T_c depends very strongly on the purity of the initial reagents in the case of TbMn_2 . In the present paper we give the results for 99.93% pure grade Tb MD terbium and 99.85% pure electrolytic manganese.

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