

MELTING DIAGRAM OF THE  $p\text{H}_2\text{-}o\text{D}_2$ ,  $p\text{H}_2\text{-HD}$ , AND  $o\text{D-HD}$  SYSTEMS

N. G. BEREZNYAK, I. V. BOGOYAVLENSKIĬ, L. V. KARNATSEVICH, and V. S. LOGAN

Physico-technical Institute, Ukrainian Academy of Sciences

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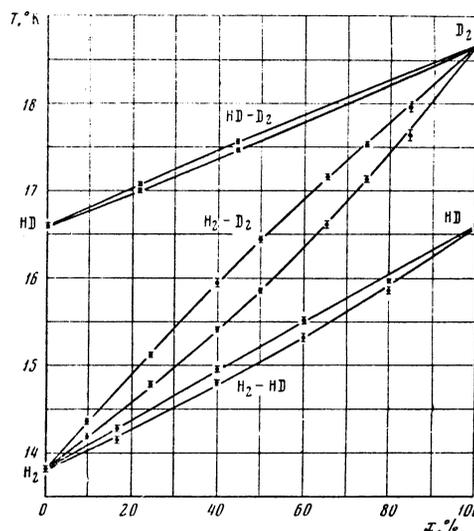
The melting diagrams of the  $\text{H}_2\text{-D}_2$ ,  $\text{H}_2\text{-HD}$ , and  $\text{D}_2\text{-HD}$  systems are investigated. The diagrams of all these systems have a cigar shape. The excess Gibbs potential  $G_e$  in the solid phase is calculated from the data for the  $\text{H}_2\text{-D}_2$  system. From the form of the dependence of  $G_e$  on the solution concentration it follows that in the solid phase near the melting diagrams the solutions are regular. The critical separation temperature calculated for  $\text{H}_2\text{-D}_2$  solutions in the solid phase is  $2.25^\circ\text{K}$ .

THE solid-liquid phase diagram was investigated earlier for the system  $\text{H}_2\text{-D}_2$ <sup>[1-3]</sup>. However, the information concerning its character is contradictory: in<sup>[1]</sup>, a peritectic diagram was observed at  $16.4^\circ\text{K}$ , whereas from<sup>[2,3]</sup>, it follows that the diagram is a cigar. In addition, an estimate of the critical lamination temperature  $T_c$  in the solid phase, which is based on the constructed diagrams, leads to great uncertainty. The values of  $T_c$  lie between 2 and  $6^\circ\text{K}$ . As noted by the authors themselves<sup>[3]</sup>, this is connected with the appreciable errors in the measurement of the solidus line, due to the slow establishment of equilibrium in the bulky sample.

The present investigation was undertaken in order to construct the melting diagrams of all binary systems made up of the hydrogen isotopes  $\text{H}_2$ ,  $\text{D}_2$ , and  $\text{HD}$ , including a more exact determination of the topology of the  $\text{H}_2\text{-D}_2$  diagram. The obtained experimental data on the melting diagrams together with the published data concerning the excess thermodynamic quantities of the liquid solution<sup>[4-6]</sup> were used to calculate these quantities in solid solutions, and also to estimate the critical lamination temperature of these solutions  $T_c$  in the solid phase.

The liquid-crystal equilibrium diagrams were investigated by the thermal-analysis method, i.e., by plotting thermograms (the dependences of the temperature of the solution sample on the time during the course of heating or cooling). To eliminate the errors connected with the lack of equilibrium, which is customary in the case of bulky samples, the cavity of the vessel, in which the investigated solution was condensed, was filled with a porous block pressed of copper powder and then sintered. By the same token, the region of crystallization of the solution was limited to pores with an average duration of approximately  $10\ \mu$ . This ensured rapid establishment of thermal and concentration equilibrium in the process of plotting the thermograms, and consequently made it possible to measure with greater accuracy the true position of the liquidus and solidus lines on the diagrams.

For most mixtures, the thermograms were plotted in the temperature interval from  $10$  to  $20^\circ\text{K}$ . The mixture  $p\text{H}_2\text{-}o\text{D}_2$  with concentration 50% was investigated



Melting diagrams of hydrogen isotope systems.

in a still wider temperature interval, from  $4.2$  to  $20^\circ\text{K}$ . No irregularities were observed on the thermograms, with the exception of the points corresponding to the start and end of the melting of the solutions.

The figure shows the obtained diagrams. As seen, they represent smooth cigars without peritectic or any other singularities, corresponding to the existence of a continuous series of solid solutions of the hydrogen isotope, at least near the melting diagrams. The calculation of the excess Gibbs potential  $G_e$  for the solid phase of  $p\text{H}_2\text{-}o\text{D}_2$  solutions gives the following dependence of  $G_e$  along the phase diagram of the fusion:

$$G_e = ax_{\text{H}_2}(1 - x_{\text{H}_2}),$$

where  $a = 9.1 \pm 0.1$  cal/mole and does not depend on the temperature, and  $x_{\text{H}_2}$  is the hydrogen concentration. This means that the solid solutions are regular, at any rate near the melting diagram. If we assume that the regularity of the solution is retained also for lower temperatures, then we can estimate the  $T_c$  of the lamination, which for the  $p\text{H}_2\text{-}o\text{D}_2$  system turns out to be

$$T_c = a/2R = 2.25^\circ\text{K},$$

where  $R$  is the gas constant.

A complete calculation of all the diagrams and a

<sup>1)</sup>We investigated modifications in equilibrium with respect to the ortho and para compositions at  $20^\circ\text{K}$ , and designated here  $p\text{H}_2$  and  $o\text{D}_2$ .

comparison of the data obtained from it with the prediction of the theoretical paper<sup>[7,8]</sup> will be performed in the future.

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